



Universitat Autònoma
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Doping and sensing in Si nanowires

X. Cartoixà, A. Miranda-Durán, R. Rurali

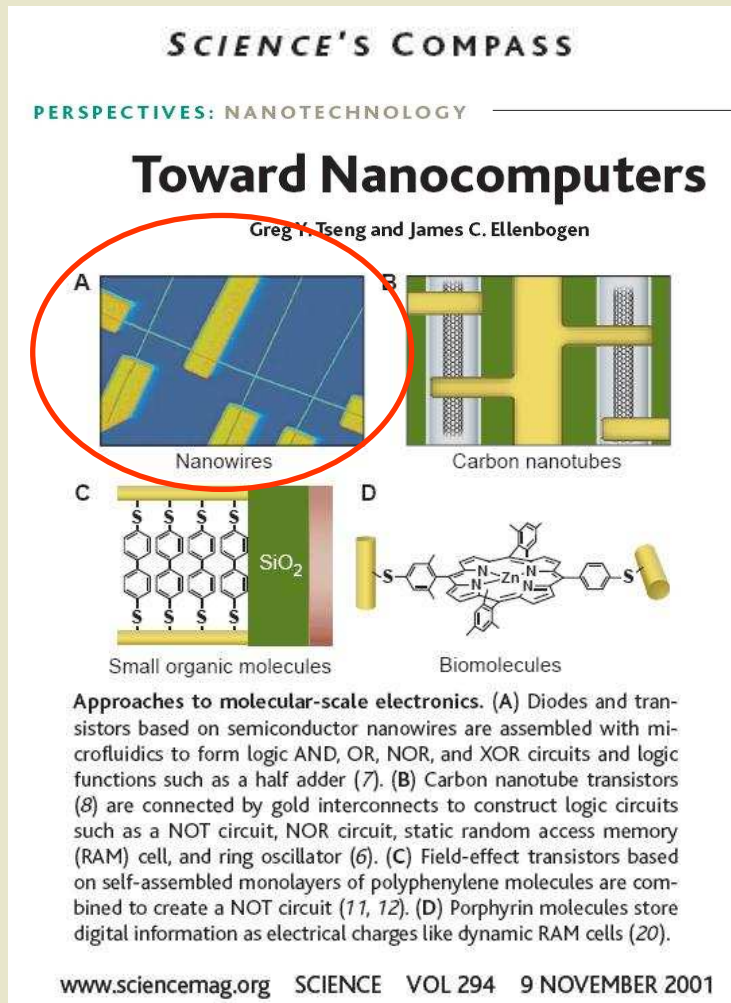
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Institut de Ciència de Materials de Barcelona - CSIC

Summary

- Why nanowires?
- Substitutional dopants in NWs
- Molecular doping and gas sensing
- Conclusions

Beyond CMOS – Emerging devices



Common paradigm

Complementing lithographic techniques with **bottom-up fabrication** and **molecular self-assembly**

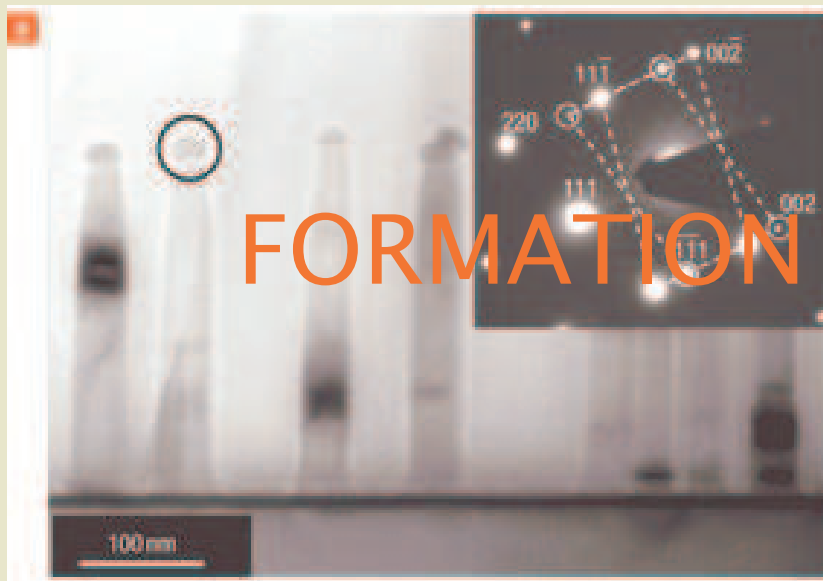
Advantages of Si nanowires

Good control of the electronic properties (always **semiconducting**)

Ideal **integration** with the existing Si technology

Why impurities?

- Can we dope nanowires effectively?
- Is any VLS/VSS catalyst cleaner than the others?
- Will dopants segregate towards the surface?
- How soluble is P in a SiNW?



FORMATION ENERGIES!!!



S. Hofmann et al., Nature Mat. 7, 372 (2008)

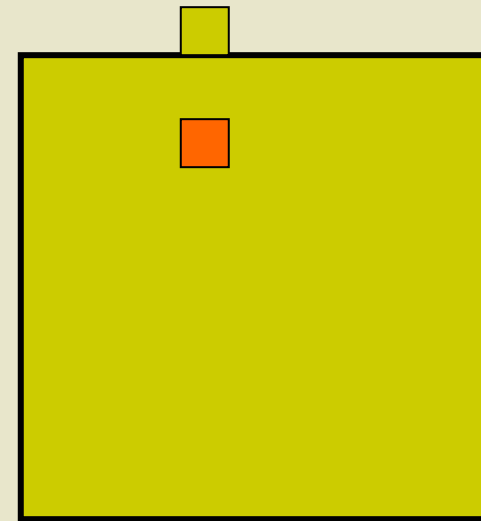
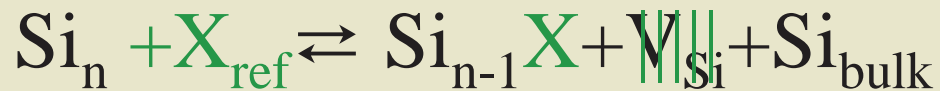
Y. Wang et al., Nature Nanotech. 1, 186 (2006)

Summary

- Why nanowires?
- **Substitutional dopants in NWs**
- Molecular doping and gas sensing
- Conclusions

ZN formalism: Dopants

- Consider the “chemical reaction” for substitutional formation



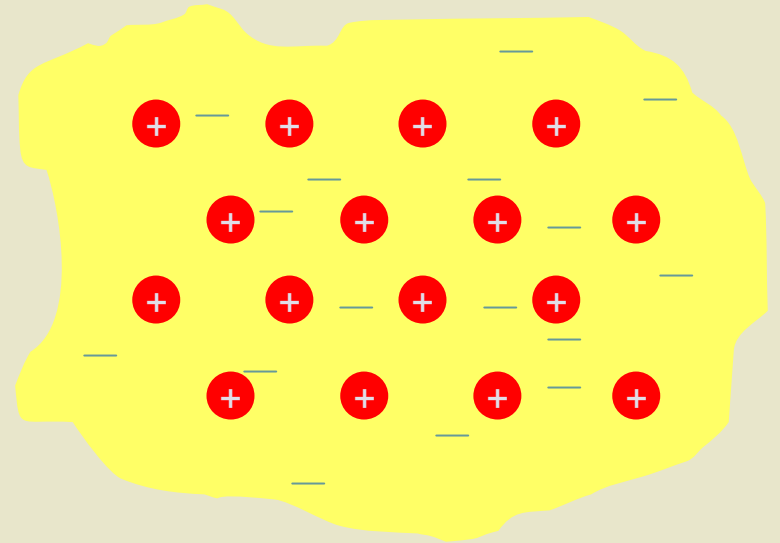
$$\Delta E_f = E_S + \mu_{\text{Si}} - E_{\text{clean}} - \mu_X = E_S - E_{\text{clean}} - \sum_i n_i \mu_i$$

S. B. Zhang and John E. Northrup,
Phys. Rev. Lett. **67**, 2339 (1991)

n_i : Number of atoms *added* into the defect

Charged cells in PBCs

- The spurious contribution due to the interaction of charge with its images must be subtracted.
- A periodic array of charges has a divergent Coulomb energy...
- ... but it can be embedded in a neutralizing jellium to make the sum convergent.
- We want the $L \rightarrow \infty$ limit.



Charged cells in PBCs: Ewald sum

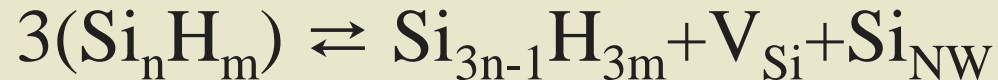
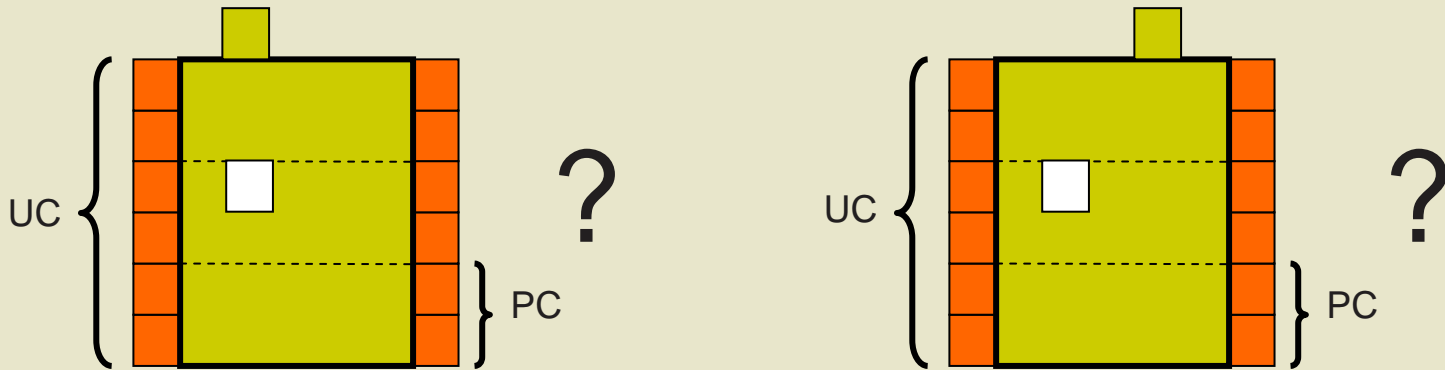
- The spurious energy is given by

$$\frac{E}{N} = \frac{1}{2} \sum_i \frac{q^2}{\epsilon R_i} = \frac{1}{2} \alpha \frac{q^2}{\epsilon L}$$

- α is the Madelung constant

$$\frac{\alpha}{L} = \sum_i \frac{\text{erfc}(\gamma R_i)}{R_i} + \sum_i \frac{4\pi}{V_c} \frac{\exp(-G_i^2 / 4\gamma^2)}{G_i^2} - \frac{2\gamma}{\sqrt{\pi}} - \frac{\pi}{V_c \gamma^2}$$

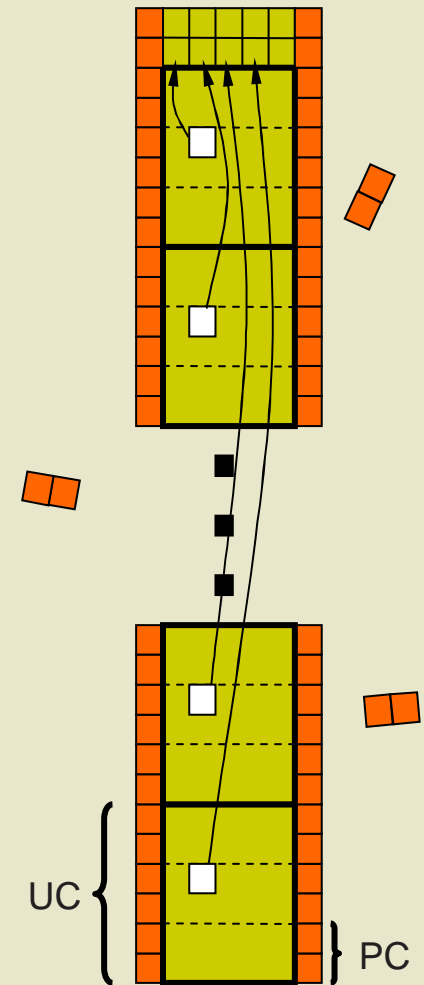
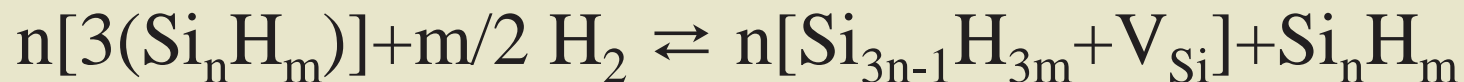
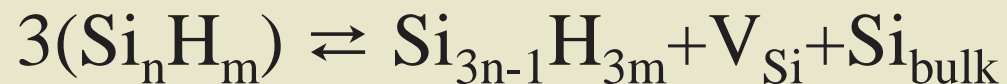
E_f in NWs: ZN formalism is hard



- In a vacancy/substitutional, you want to add the displaced atom to the NW, but...
- ... where does the displaced atom go?
- ... what's the energy of that atom?
- ... what about surface passivation?

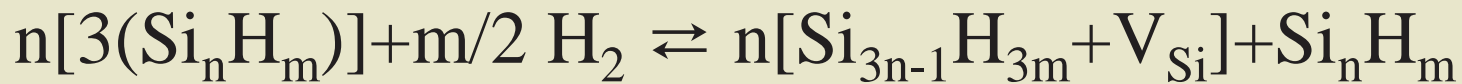
E_f in NWs: Consider many instances!

- We consider the creation of as many defects as necessary to form a new primitive cell.
- Everything can be readily computed from first-principles.
- Reaction tells you how to deal with surface passivation.



E_f in NWs: Modified expression

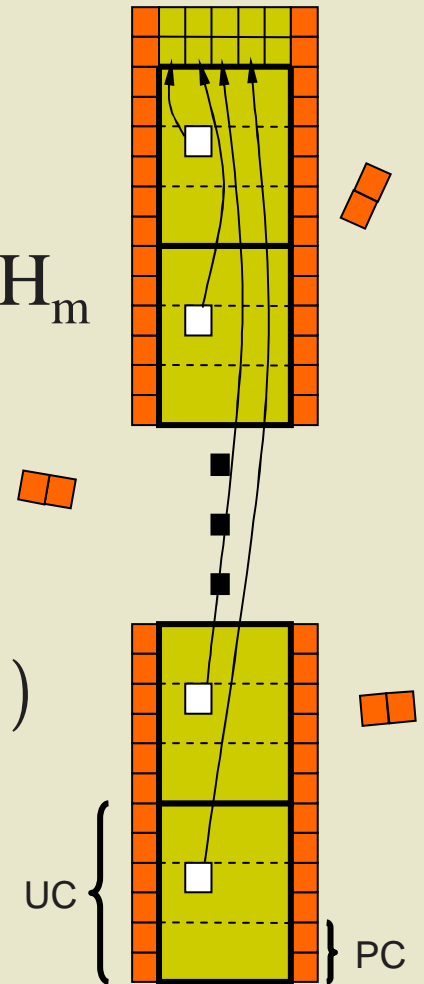
- The formation energy corresponding to the reaction



is

$$\Delta E^f = E_{tot}^D - NE_{NW}^{PC} - \frac{n^{Si}}{n_{Si}^{PC}} \left(E_{NW}^{PC} - \frac{n_H^{PC}}{2} E_{H_2} \right) + q(\epsilon_v + \mu_e)$$

Rurali and Cartoixà,
Nano Lett. **9** (3), 975-979 (2009)



E_f in NWs: Net charges

- The unit cell will no longer need to be cubic, but even if we choose it to be so...
- ... ϵ is now a *tensor*
- Need to start from the potential due to a point charge embedded in an anisotropic medium:

$$V(r) = \frac{1}{\sqrt{\det(\epsilon)}} \frac{1}{\sqrt{r \cdot \epsilon^{-1} \cdot r}}$$

E_f in NWs: Ewald sum

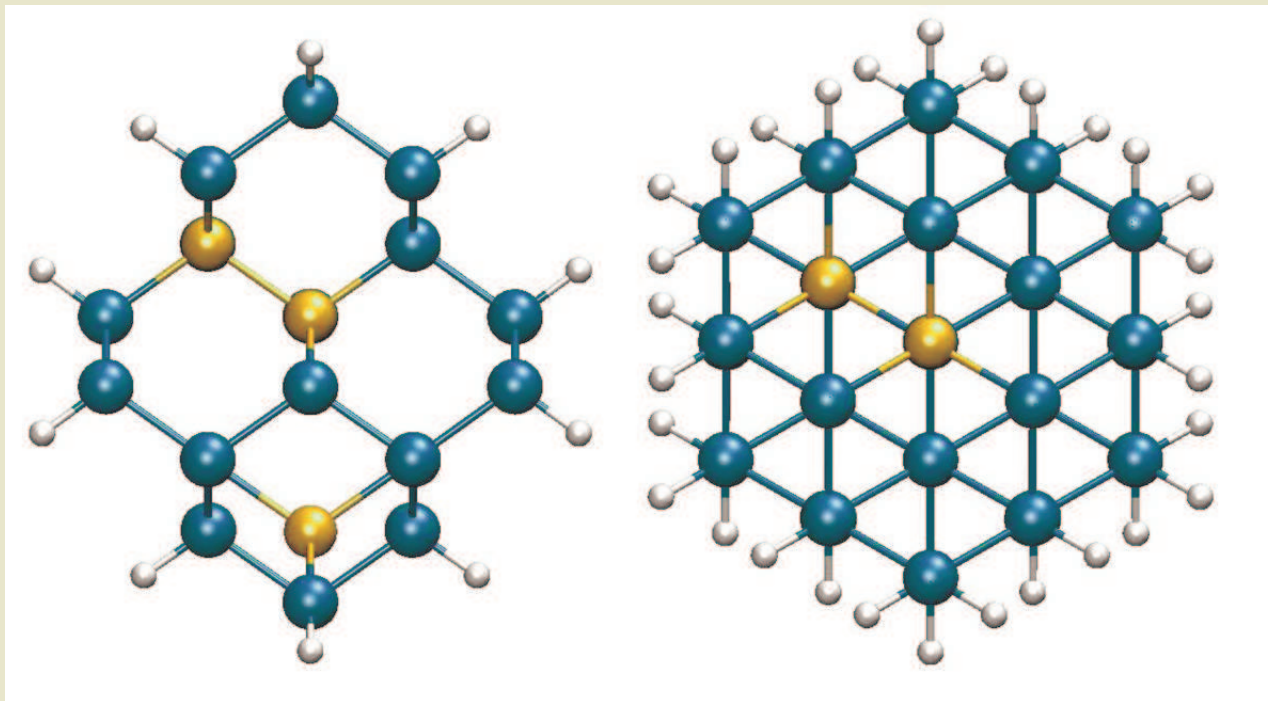
- The general expression needed to compute the Madelung constant for the NWs is

$$\frac{\alpha}{L} = \sum_i \frac{1}{\sqrt{\det(\epsilon)}} \frac{\operatorname{erfc}(\gamma \sqrt{R_i \cdot \epsilon^{-1} \cdot R_i})}{\sqrt{R_i \cdot \epsilon^{-1} \cdot R_i}} + \sum_i \frac{4\pi}{V_c} \frac{\exp(-G_i \cdot \epsilon \cdot G_i / 4\gamma^2)}{G_i \cdot \epsilon \cdot G_i} - \frac{2\gamma}{\sqrt{\pi \det(\epsilon)}} - \frac{\pi}{V_c \gamma^2}$$

Rurali and Cartoixà,
Nano Lett. **9** (3), 975-979 (2009)

Al@SiNWs: Configurations

- We have studied substitutional and interstitial Al atoms in $\varnothing=1.0$ nm and 1.5 nm silicon NWs.

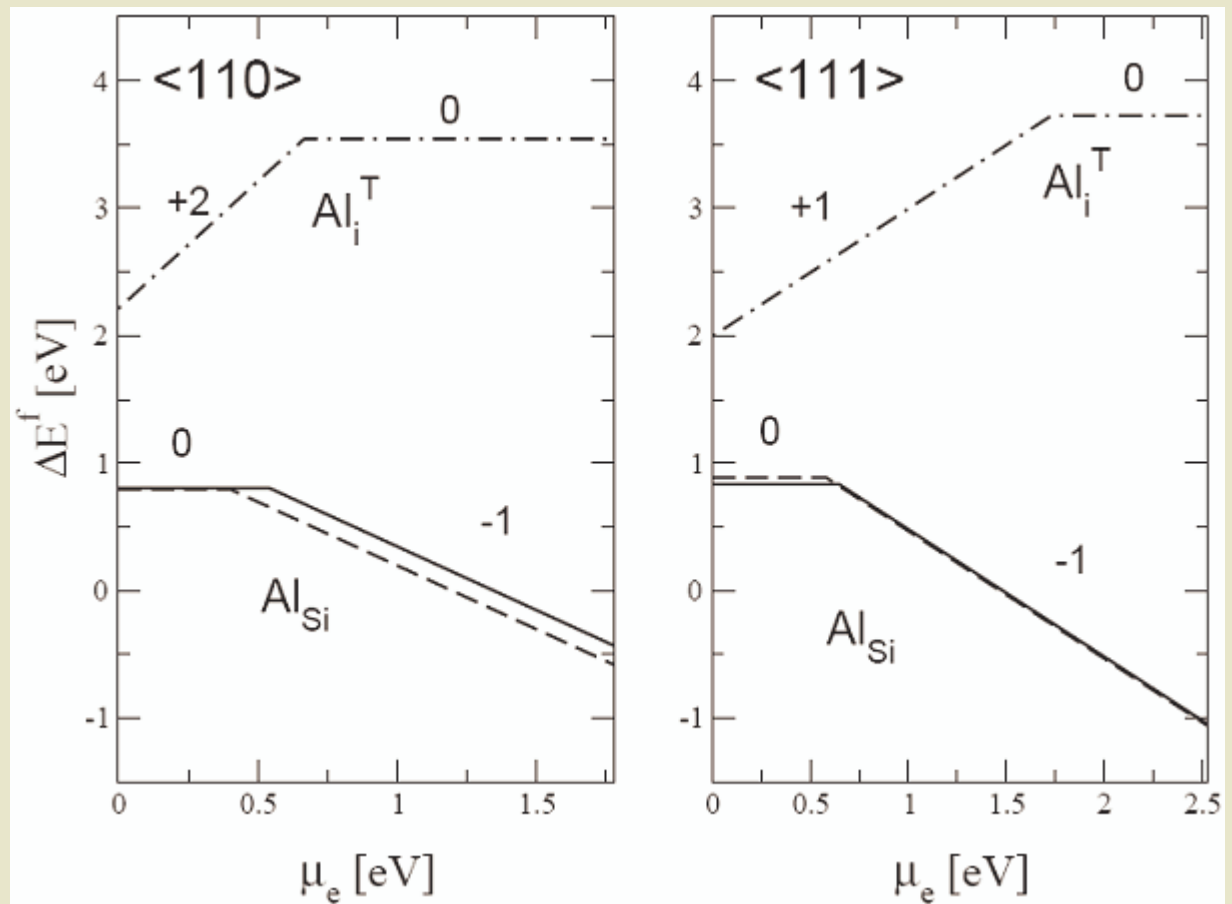


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Al@SiNWs: Configurations

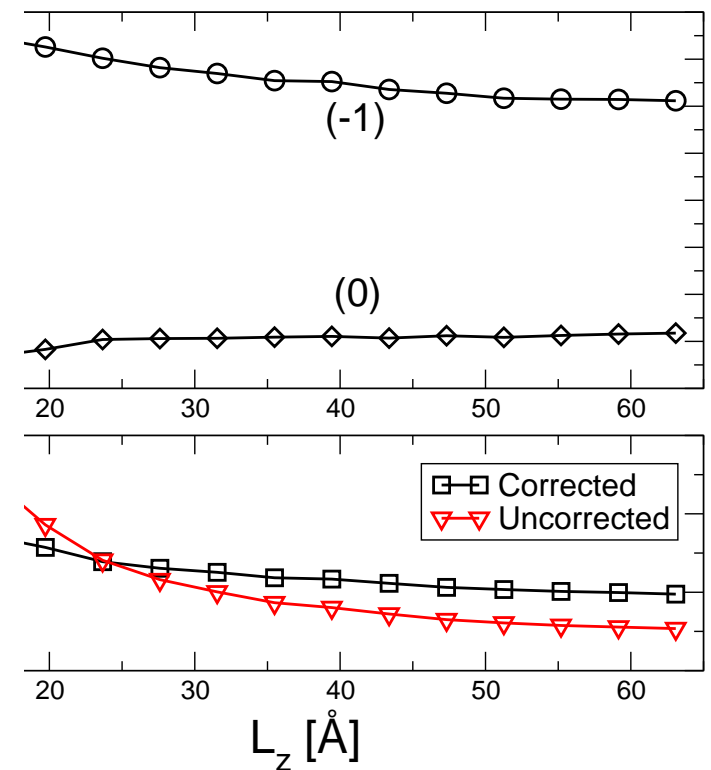
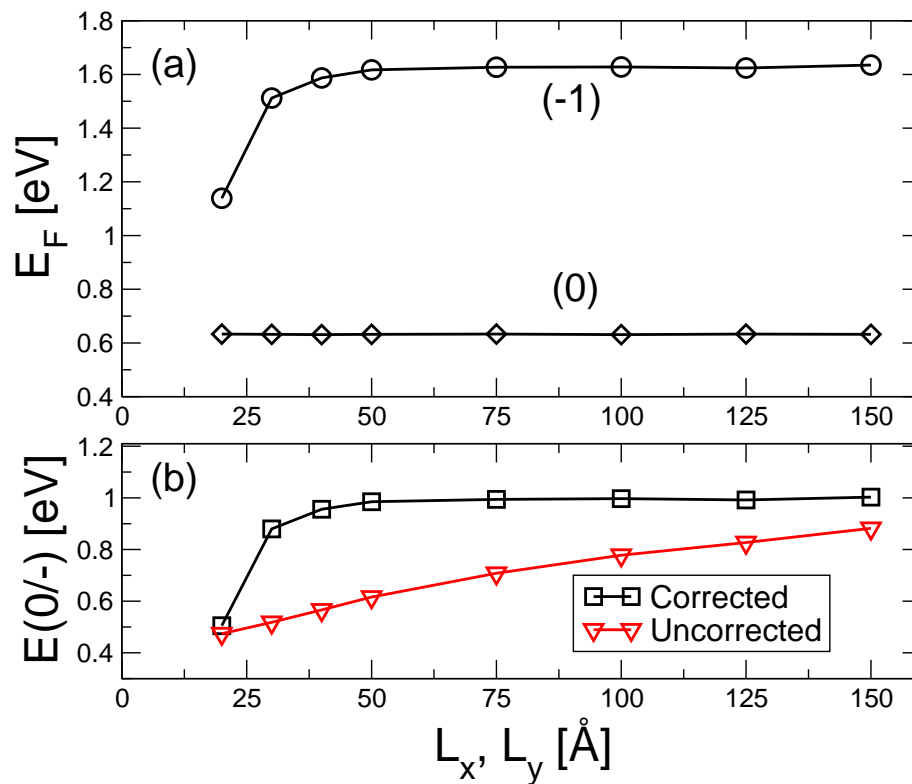
- Substitutional Al is preferred, Al is a acceptor.
- For heavily n-doped, Al dilution is eased.



Rurali and Cartoixà,
Nano Lett. **9** (3), 975-979 (2009)

Al@SiNWs: Convergence study

- Need a $50 \times 50 \times 60 \text{ \AA}^3$ supercell for a $\langle 110 \rangle$



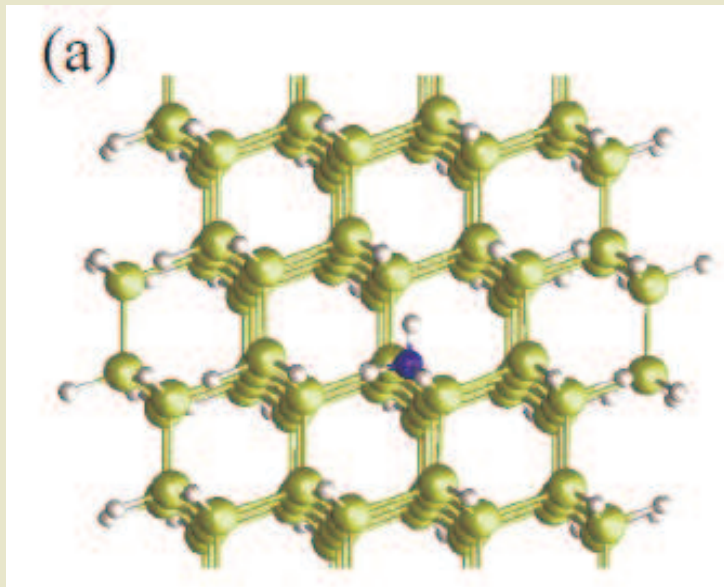
Rurali, Palumbo and Cartoixà,
Phys. Rev. B **81**, 235304 (2010)

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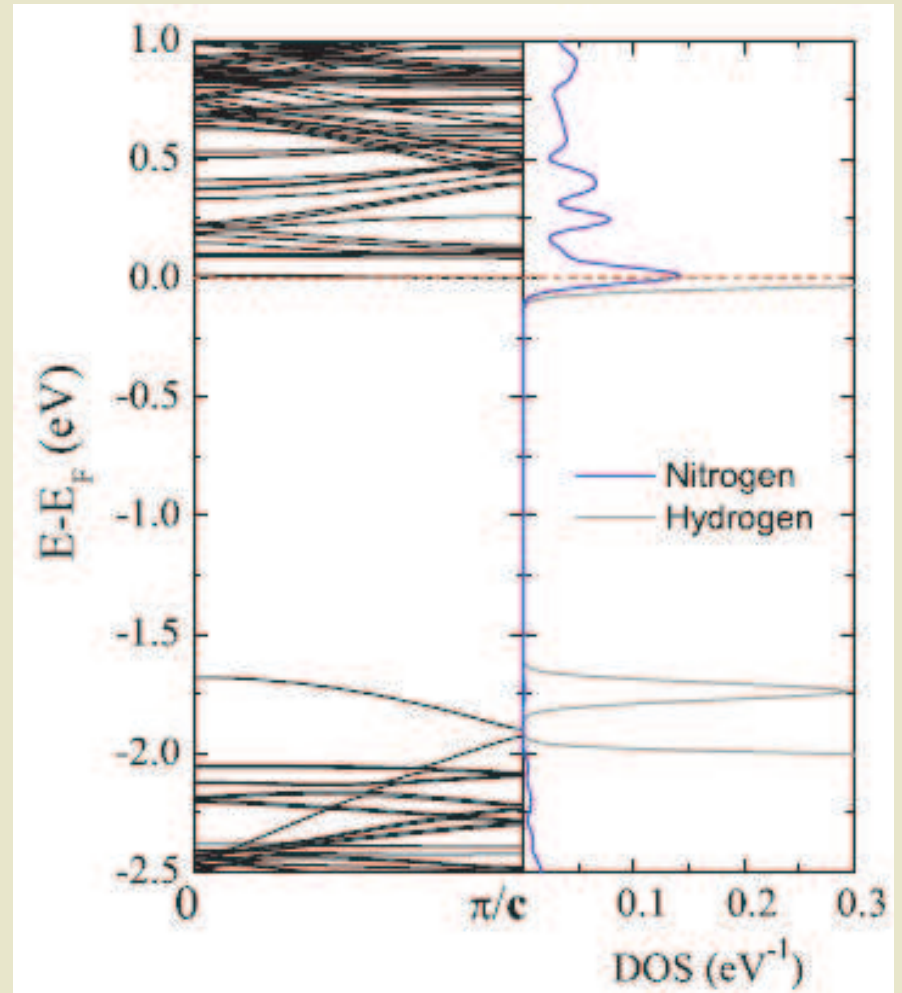
NH₃ on SiNWs

- NH₃@DB acts as a donor for SiNWs, surfaces



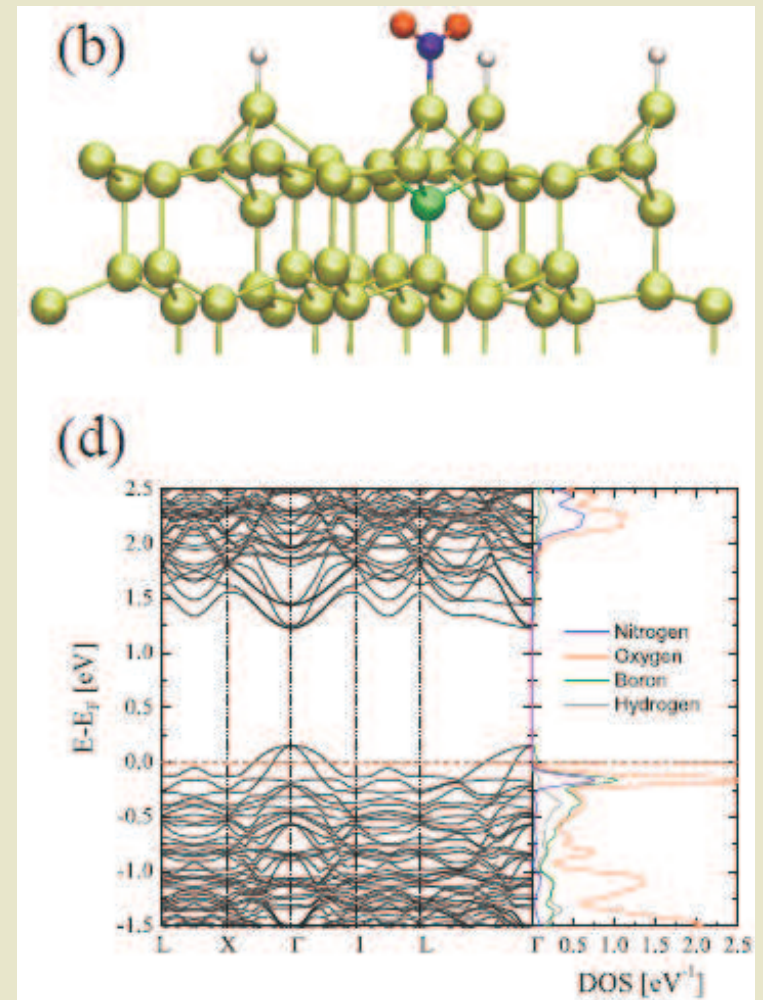
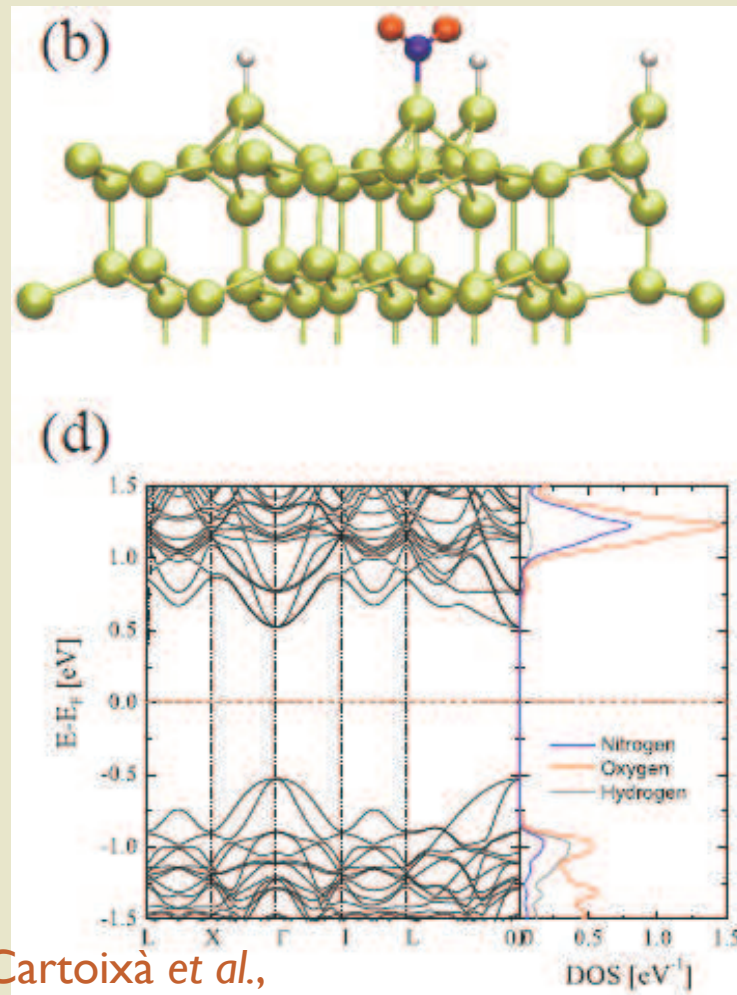
Miranda-Durán, Cartoixà *et al.*,
Nano Lett. **10**, 3590 (2010)

Also see poster 89!!



NO₂ on SiNWs

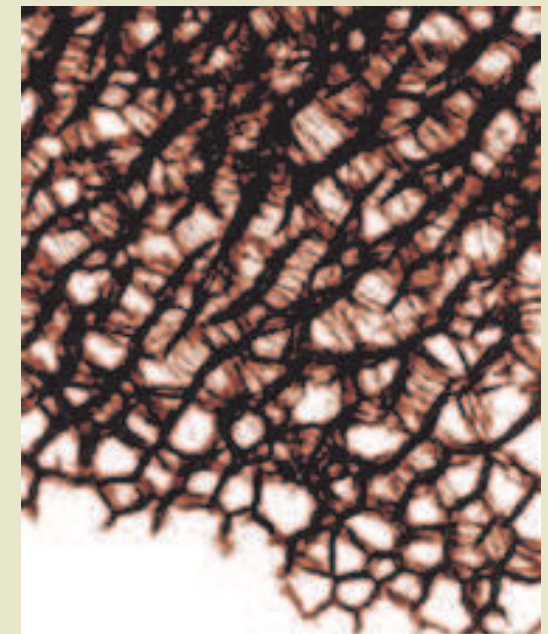
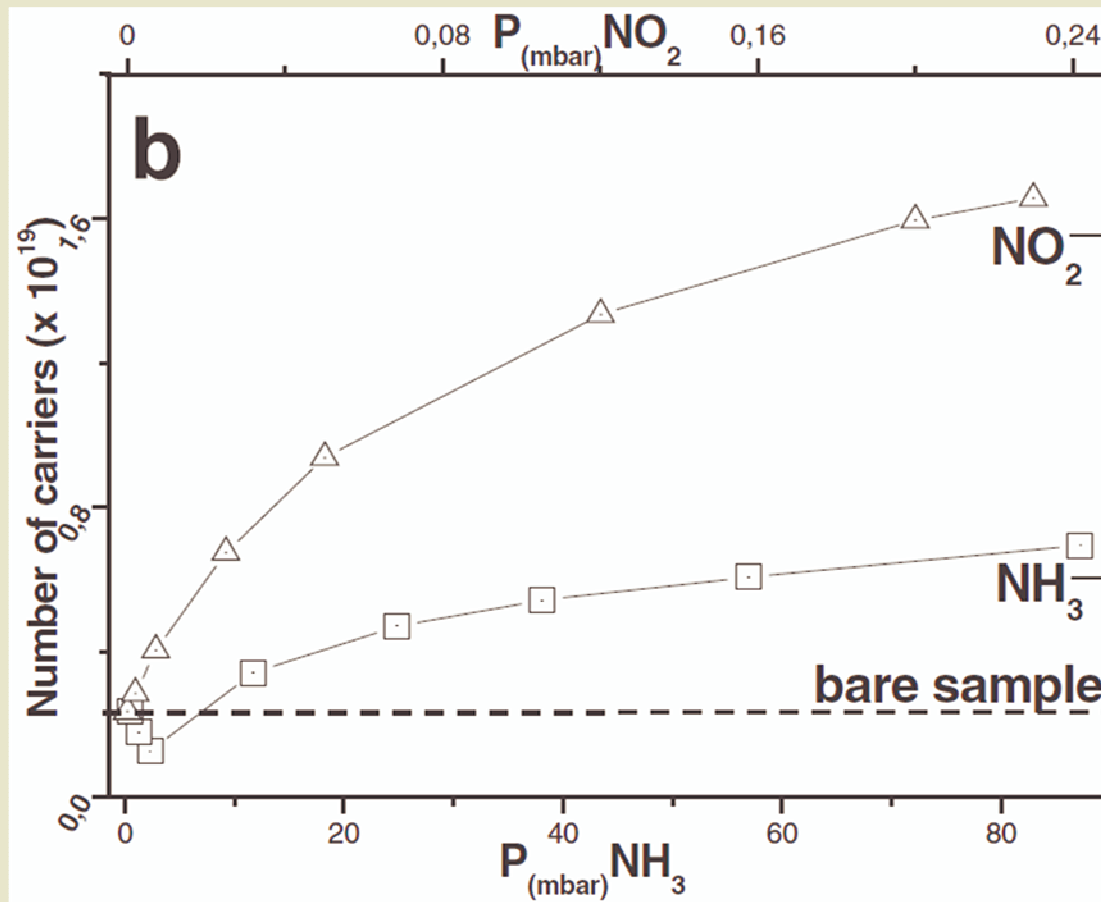
- NO₂ can reactivate a B-DB complex.



Miranda-Durán, Cartoixà *et al.*,
Nano Lett. **10**, 3590 (2010)

Experimental results in porous Si

- B stops etching
- Subsurface B are more frequent than inner B



Cullis and Canham,
Nature **353**, 335 (1991)

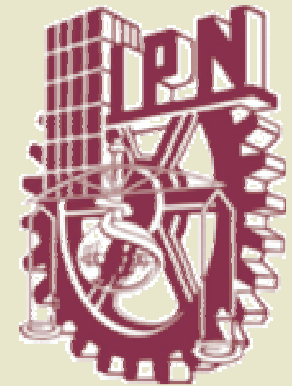
Garrone *et al.*, *Adv. Mater.*
17 (5), 3590 (2005)

Conclusions

- Substitutional impurities traditionally used for semiconductor doping have too high activation energies to be effective for ultrathin nanowires.
- Molecular adsorption can be a viable alternative to substitutional doping in thin NWs.
- The conductivity behavior of mesoporous Si under the presence of NO_2 and NH_3 has been explained.

Collaborators

- ICMAB:
 - Dr. Enric Canadell
- Instituto Politécnico Nacional (México)
 - Prof. Miguel Cruz-Irisson
- Università di Roma “Tor Vergata”
 - Dr. Maurizia Palummo
- MICINN, TEC2009-06986





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**THANK YOU FOR
YOUR ATTENTION**
